Biophysical characterization of synthetic adhesins for predicting and tuning engineered living material properties

To expand the integration of synthetic biology in material science, we quantitatively characterized a bacterial synthetic adhesion toolbox through several biophysical methods. Specifically, we measured the synthetic adhesins’ turnover rates, their surface density and spatial distribution, their membrane diffusion rate, and their intermolecular bond-breaking force. With these parameters, we then demonstrated how to predict and quantitatively tune the macroscopic properties of engineered living materials consisting of cells expressing these synthetic adhesins, which is essential for applications such as bioprinting.
Biophysical characterization of synthetic adhesins for predicting and tuning engineered living material properties

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SUMMARY
Bacterial synthetic multicellular systems are promising platforms for engineered living materials (ELMs) for medical, biosynthesis, environmental, and smart materials applications. Recent advancements in genetically encoded adhesion toolkits have enabled precise manipulation of cell-cell adhesion and the design and patterning of self-assembled multicellular materials. However, in contrast to gene regulation in synthetic biology, the characterization and control of synthetic adhesins remains limited. Here, we demonstrate the quantitative characterization of a bacterial synthetic adhesion toolbox through various biophysical methods. We determine key parameters, including number of adhesins per cell, in-membrane diffusion constant, production and decay rates, and bond-breaking force between adhesins. With these parameters, we demonstrate the bottom-up prediction and quantitative tuning of macroscopic ELM properties (tensile strength) and, furthermore, that cells inside ELMs are connected only by a small fraction of available adhesins. These results enable the rational engineering, characterization, and modeling of other synthetic and natural adhesins and multicellular consortia.

INTRODUCTION
The emerging discipline of engineered living materials (ELMs) highlights many outstanding properties over traditional materials, and synthetic biology methods are a key up-and-coming approach for ELMs.1–5 Cells promise to be versatile and modular material building blocks for ELMs that can harness the near-unlimited complexity and functionality of multicellular systems.6,6,7,12 For example, bacterial synthetic multicellular “consortia” can be used to enable programmable smart materials and artificial tissues including bioprinting (Figure 1A),1,4 modularizable pathway engineering platforms for natural drug biosynthesis,1 in vivo drug-delivery vehicles,9,10 and build-to-understand biofilm disease models,11 all while being sustainable and biodegradable.1

The control of cell-cell adhesion is fundamental for multicellular ELMs, as it enables cells to stick together, allows patterning algorithms (such as self-assembly, differentiation, and potentially synthetic development),6,6,7,12 and also determines macroscopic material properties such as viscoelasticity or tensile strength.1,13 Previous work established a synthetic cell-cell adhesion toolbox in bacteria consisting of a library of heterophilic nanobody-antigen (Nb-Ag) adhesin pairs that are surface displayed on the outer cell membrane7,14 (Figure 1B). This toolbox provides control over adhesion strength and tunability through chemical inducibility, specificity
between adhesion pairs, and homophilic cell-cell interactions between cells due to composability. ELMs at various length scales with interesting properties such as self-growth, programmability, fast recovery under stretching or bending, and open-surface microfluidics have already been demonstrated with this synthetic adhesin toolbox. Additionally, other synthetic cell-cell adhesins have been developed in both bacterial and eukaryotic systems.

The quantitative and biophysical characterization of such synthetic adhesion toolboxes is significantly lacking, but it is key for the bottom-up prediction of macroscopic material properties from the synthetic building parts, which is also deficient. This is in stark contrast to synthetic biology at the molecular and gene-regulatory level, where, for example, synthetic promoter and repressor systems have been characterized in great detail and successfully used to quantitatively predict the behaviors and functionalities of complex genetic circuits. Synthetic biology pursues the ability to flexibly engineer many-component systems from libraries of standardized and well-characterized parts; consequently, the equivalent characterization of synthetic adhesins for ELMs is key.

This paper presents a general framework for quantifying the key parameters of synthetic adhesins in Escherichia coli (Figure 1C) that are crucial for engineering synthetic multicellular materials. Specifically, we measure the synthetic adhesins’ production and decay rate, their density, their spatial distribution and diffusion inside the cell membrane, and their intermolecular bond-breaking force (Figure 1C). For example, the tensile strength and the viscoelasticity of an ELM may depend on whether cells can slide past each other and rearrange their relative position and orientation, which critically depends on the number of adhesins per cell, their stability (longevity), their pairwise binding strength (bond-breaking force), and their ability to move within the cellular membrane. Based on these measured parameters, we then demonstrate how to predict bottom-up and quantitatively tune the macroscopic mechanical properties of an ELM. As a suitable test case, we focus on material tensile strength, which is relevant for bioprinting among other applications (Figure 1A). This generic framework for quantifying synthetic cell adhesin properties will also support basic science and engineering beyond ELMs.

Figure 1. The quantification of the main biophysical parameters for synthetic adhesins is the key for the rational bottom-up engineering of living bacterial materials including the quantitative and predictable tuning of the material properties (A) Schematic of engineered living materials (ELMs) made from E. coli bacteria mediated by synthetic adhesins with utility, for example, for 3D bioprinting. (B) Schematic of previously engineered synthetic cell-cell adhesin toolbox. (C) Key biophysical parameters of synthetic adhesins in E. coli are determined (see Table 1 for measured values). Illustrations partially adapted from Glass and Riedel-Kruse.

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RESULTS

Number of adhesins and spatial surface distribution

First, we quantified the number of adhesins per E. coli cell by expressing Nb against GFP at maximum induction (300 ng/mL anhydrous tetracycline [aTc]) on these cells and then comparing the fluorescence intensity against a known standard. Specifically, we labeled the cells with GFP and compared the resulting fluorescence intensity per cell with a non-fluorescent cell bathed in a solution of known GFP concentration (365 nM) (Figure 2A and Note S1). We corrected for background and out-of-focus-plane contributions, similar to previous work (Figure 2B). As 1 nM GFP solution corresponds to 0.6 molecules/μm², the number of adhesin molecules per cell under maximum induction then is \( N_{adh}(300) = 15,300 \pm 4,100 \) (mean ± SEM, henceforth used throughout unless stated otherwise) (n = 20 cells). As GFP is only labeling the bacterial surface, this corresponds to a surface density of \( \rho_{adh} = 3,400 \pm 1,200 \) adhesins/μm² given a bacterial cell length and radius of \( l = 1.80 \pm 0.25 \) μm and \( r = 0.40 \pm 0.10 \) μm, respectively (Note S1). At an intermediate induction of 100 ng/mL aTc—as in many of the following experiments—this would then
correspond to $7,300 \pm 1,800$ adhesins per cell, or a surface density of $1,700 \pm 600$. A potential caveat with this method is that in solution GFP can exist in a bright monomeric and a dark oligomeric form, which could lead to an overestimation of the adhesin number on the cell surface, as bound GFP are in the monomeric form. Yet our result is in line with previous work that used a western blotting technique (Note S1).

Next, we determined the dependencies of the total adhesin number on aTc inducer concentration (Figure 2C). We tested aTc concentrations between 0 and 1,000 ng/mL. We found effective saturation at 300 ng/mL, and we observed unhealthy characteristics at 1,000 ng/mL (but not below 600 ng/mL), e.g., filamentous growth or lysis (data not shown). This saturation concentration is consistent with prior work on the TetR repressor system. We then fitted the following Hill equation (Note S1 and Figure 2C):

$$I(C) = f \frac{N_{adh,max}}{N_{pix}} \frac{C^n}{C_{1/2}^n + C^n},$$

(Equation 1)

with $C$ being the inducer concentration, $I(C)$ the corresponding fluorescence intensity per pixel, $N_{pix}$ the total number of pixels per cell, $C_{1/2} = 85.6 \pm 6.5$ ng/mL (the inducer concentration for half-maximal expression), $n = 1.8 \pm 0.1$ (the Hill coefficient), which is consistent with a cooperativity factor of 2 for aTc, being a conversion factor between the adhesin number and fluorescence intensity, and $N_{adh,max} = 17,400 \pm 9,600$ molecules/cell (which is close to $N_{adh}(300)$).

Next, we investigated whether adhesins are homogeneously distributed over the surface or partially localized. We selected cells with both poles in focus and measured the fluorescence intensities along the circumference of the cell, at the poles, and in the middle (Figure 2D). We observed that the fluorescence is normally distributed along this circumference, where the mean and standard deviation of 41 ± 4 a.u./pixel is consistent with the total number of adhesins per cell as determined above (Figure 2E and Note S1). Next, we measured the fluorescence intensity of 10 pixels in the middle and at one of the poles, and we did not observe any significant difference (Student’s t test, $p = 0.56$). Thus, we conclude that the adhesins are homogeneously distributed over the cell surface.

**Kinetics of adhesin turnover**

Next, we were interested in characterizing the kinetics of the adhesin number at a given inducer concentration, $N_{adh}(C)$, due to their effective production rate $a(C)$ and their effective decay rate $b_{eff}$, with

$$a(C) = a_{max} \frac{C^n}{C_{1/2}^n + C^n} = N_{adh}(C) \cdot b_{eff}$$

(Equation 2)

and where $a_{max}$ is the maximum protein production rate. Generally, $b_{eff}$ is the sum of specific degradation rate $b_{deg}$ due to destruction and the dilution rate $b_{div}$ due to cell division. The specific degradation rate also strongly depends on environmental conditions, for example, growth vs. starvation, where degradation machinery has different activities.

To determine these rates under growth conditions, we set up an overnight culture of uninduced GFP-Nb-expressing bacterial cells, which were later back-diluted for 4 h and then transferred in medium with 100 ng/mL aTc (Figure 3A). We then took samples at different time points $t$, labeled them with GFP, and quantified adhesin expression with fluorescent microscopy (Figures 3A and 3B). We fitted an exponential growth function for the average intensity per pixel $I(t)$ (Note S2):
\[ I(t) = f \cdot \frac{a_{\text{max}}}{b_{\text{eff}}} \cdot \frac{C_n}{C_{1/2} + C_n} \cdot (1 - e^{-t \cdot b_{\text{eff}}}). \]  

(Equation 3)

Using \( f \), \( C_{1/2} \), and \( n \) as above, the fit to Equation 3 leads to \( a_{\text{max}} = 9,900 \pm 2,100 \) adhesins/cell/h (for example, at 100 ng/mL aTc this corresponds to \( a(100) = 5.600 \pm 1.200 \) adhesins/cell/h, and which is ultimately also promoter and plasmid specific) and \( b_{\text{eff}} = 0.78 \pm 0.12 \, \text{h}^{-1} \) (\( n = 6 \) samples). (Figure 3B). This effective decay rate is not statistically different from the dilution rate due to cell-division rate of \( 0.82 \pm 0.30 \, \text{h}^{-1} \) (\( n = 8 \) samples) as we determined from OD\(_{600}\) measurements of the cell culture (Note S2, Student’s t test, \( p = 0.6 \)), i.e., dilution dominates over degradation. Both decay and production rates are consistent with literature values in related systems.

To measure the specific degradation rate of adhesins during starvation conditions, we induced adhesion expression followed by labeling with GFP, then transferred these cells onto agarose pads without nutrients and without inducer (Figure 3C). Using quantitative fluorescence time-lapse microscopy, we monitored the average fluorescence intensity per pixel over time (Figure 3D). We fitted an exponential decay function, where now \( a \) and \( b_{\text{eff}} \) are zero as the cells are not dividing (Figure 3E). We also corrected for potential photofading and active GFP degradation, leading to...
$b_{\text{deg}} = 0.05 \pm 0.01 \text{ h}^{-1} (n = 16)$ (Note S2), which is consistent to degradation rates measured for other E. coli proteins.$^{39}$ We expect the mechanisms of degradation to include outer membrane proteases$^{39-41}$ or periplasmic proteases,$^{42}$ as has been elucidated for several outer membrane proteins in E. coli, rather than environmental factors (i.e., extreme temperatures or pH).

**Diffusion coefficient**

We determined the lateral diffusion constant of adhesins in the outer cell membrane in vivo through fluorescence recovery after photobleaching (FRAP) on a confocal microscope.$^{43}$ First, we chose a bleaching location at three-quarters of the cell length, and we determined the mobile fraction of $M_f = 0.37 \pm 0.04$ with a diffusion coefficient of $D = 0.36 \pm 0.06 \mu m^2/s (n = 16$ cells) (Figures 4A and 4B). We controlled these measurements for various instrument settings (Note S3). Furthermore, these values are consistent with measurements on other outer membrane proteins (OMPs) in E. coli with typical values between 0.05 and 0.6 $\mu m^2/s$.$^{44}$ We also expected that the adhesin LysM domain for peptidoglycan binding would not affect the diffusion dynamics based on results in previous studies.$^{45,46}$

We next investigated whether this diffusion constant is dependent on the region of the cell. Consequently, we measured diffusion constants at the center, at three-quarters of the cell length, and at the pole of $0.37 \pm 0.05 \mu m^2/s, 0.36 \pm 0.06 \mu m^2/s,$ and $0.35 \pm 0.06 \mu m^2/s,$ respectively (Student’s t test, $p > 0.5$) (Figures 4C and 4D). Note that the raw value at the pole was multiplied by 2 in order to correct for the fact...
that at this position the fluorescence influx during recovery only proceeds from one direction. Due to OMPs being organized in islands, there might be other diffusion kinetics on significantly different time and length scales but that are outside of the present scope. We conclude that the lateral diffusion of the adhesin molecules in the cell membrane is the same everywhere on the cell surface.

Adhesin bond-breaking force

We used optical trapping experiments to measure the force necessary to break the molecular bond between a pair of complementary Nb-Ag adhesins (Figure 5A). We induced cells to express EPEA Nb (EPEA refers to the 4 amino acids of this small peptide, i.e., glutamic acid, proline, glutamic acid, alanine) and functionalized streptavidin-coated polystyrene beads (diameter 1.7 μm) to present EPEA Ag-linker-biotin-streptavidin on their surface (experimental procedures). We introduced these functionalized beads into a custom-made flow cell and captured an individual bead with the optical trapping laser. We then brought the bead into contact with an E. coli cell that was unspecifically bound by poly-L-lysine (PLL) to the glass surface of the flow cell. After confirming contact between bead and cell, we oscillated the
optical trapping laser back and forth with frequencies and amplitudes in the ranges of \( f = 0.01 - 0.22 \, \text{Hz} \) and \( A = 2.2 \, \mu\text{m} \), respectively. This led to frequent binding and subsequent breaking events between the bead and the cell, as recorded by the displacement between bead and laser, which—when multiplied by the calibrated stiffness of the optical trap—reveals the corresponding breaking forces (experimental procedures) (Figure 5B).

Zero breaking force implies that no attachment had formed between bead and cell prior to pulling. For the majority of the 282 trials conducted, we used a loading rate of 20 pN/s; however, we investigated a range of loading rates between 5 pN/s and 100 pN/s, finding no statistically significant variation in Nb-Ag bond rupture force.

We observed a distribution of breaking forces for different levels of aTc induction (Figure 5C). Note that forces above \( \sim 40 \, \text{pN} \) are not considered valid due to equipment limitations, as the force-displacement relationship for the optical trap becomes non-linear, which does not affect reliability of the reported final results (experimental procedures). We did not observe any binding at 0 ng/mL aTc induction, implying that the cell-bead attachment is adhesin specific (Figure 5C). Given the density of potential binding sites observed in fluorescence measurements above, we estimated 30 ng/mL aTc to form between 0 and 2 adhesin bonds given the dimensions of the cell, bead, and length of the entire adhesin bond (including linkers) (Figure 2 and Note S4). We then hypothesized that this force distribution (Figure 5C) clusters around multiples of approximately 16 pN, a value that would correspond to the single bond rupturing force. Consequently, we fitted a sum of two Gaussian distributions centered on multiples of an unknown value and obtained a breaking force of \( F_b = 16.1 \pm 0.4 \, \text{pN} \) (\( n = 14 \) cells) (Figure 5D); in contrast, fitting to a single or three Gaussians led to poor fits (Note S4). In addition, we determined that multiple pulling trials on the same cell did not lead to a significant destruction of the adhesins, illustrated by the cell’s ability to keep forming bonds with the bead after several binding/unbinding events (Note S4). We also attempted to determine the binding kinetics \( k_{on} \) and \( k_{off} \), similar to data reported in El-Kirat-Chatel et al.\(^{51}\) Given our instrument resolution, we can only provide the lower bound for \( k_{on} \) to be \( 1.5 \, \text{M}^{-1} \text{s}^{-1} \), while \( k_{off} \) could not be distinguished to be different from 0 (Note S4).

Note that different Ag-Nb pairs are expected to have different bond-breaking forces in general. Based on the molecular interactions involved (i.e., electrostatic interactions, hydrogen bonds, van der Waals forces, and hydrophobic interactions), bond-breaking forces are expected to be on the order of tens of pN; for example, atomic force microscopy reveals the rupture force for the Nb-GFP complex \textit{in vitro}\(^{52}\) in the range of 28–56 pN, or for natural adhesins, such as cadherins, the rupture varies between 35 and 150 pN.\(^{53}\) Furthermore, these values can be modulated by changing the Ag-Nb affinities, for example due to point mutations that determine the structure of the binding sites\(^{54}\) or through the development of fusion nanobodies.\(^{55}\) Other engineered systems that promote cell-cell binding, such as the Spycatcher-Spytag system with its covalent interactions, present breaking forces above 1 nN.\(^{56}\)

**Tuning ELM properties: Tensile strength**

We next investigated whether macroscopic material properties could be quantitatively tuned and predicted from these molecular and cellular parameters. We chose tensile strength as a suitable test bed, which is defined as the maximum stress (i.e., pulling force per cross-sectional area) that a material can withstand before breaking. We used the previously developed homophilic cells,\(^{7}\) i.e., cells expressing both Ag and Nb adhesin of the same p53TA pair, which leads to a homogeneous material
of one cell type and avoids mixing issues when working with the two strains, each just expressing one of the corresponding adhesins (Note S5). We prepared high-cell-density pellets through centrifugation inside a syringe, which were then extruded through a blunt needle with a syringe pump that was kept at a constant flow rate. The extruded material string eventually ruptures under its own weight (Figures 6A and 6B).

This rupture point allowed us to experimentally determine the tensile strength \( \sigma_{\text{exp}} = \frac{mg}{A_s l_s} \), with the gravitational force \( mg \), the cross-sectional area \( A_s \), the length \( l_s \), and the volume \( V_s = l_s A_s \) of extruded material string below the breaking point, the material density \( \rho = m/V_s = 1.100 \pm 0.050 \text{ kg/m}^3 \), and the standard gravity \( g = 9.81 \text{ m/s}^2 \). This then led to tensile strengths \( \sigma_{\text{exp}}(C) \) in the range of to 1.37 – 3.53 kPa, depending on induction level \( C \) (Figure 6C). The fact that the tensile strength at zero induction level, \( \sigma_{\text{exp}}(0) \), is not zero can be understood from other (potentially non-specific) adhesive interactions between cells. The difference in tensile strength between non-induced and induced cells of \( \delta_{\text{exp}}(C) = \sigma_{\text{exp}}(C) - \sigma_{\text{exp}}(0) \) is then due to synthetic adhesin pairs. Under the present experimental conditions, this “excess tensile strength” \( \delta_{\text{exp}}(C) \) also appears to be proportional to the synthetic adhesin expression level according to Equation 1 (Figure 6C and Note S5). We thus observed a systematic and significant increase in string length and, hence, tensile strength between non-induced, partially induced, and fully induced adhesin levels (Figure 6C), highlighting the tunability of this material property.

**Quantitative bottom-up material property predictions**

Finally, we investigated whether this excess tensile strength, \( \delta(C) \), could be predicted from first principles and quantitatively based on the earlier measured molecular and cellular parameters. \( \delta(C) \) depends on the number of adhesin bonds between neighboring cells that need to rupture, which in turn depends on the area of contact between any two cells as well as the overall ordering of cell packing. We performed confocal imaging of the material string to investigate how cells are packed and ordered after having been extruded. We observed dense packing, i.e., a solid-to-volume fraction of \( f_v = 0.5 \) (also confirmed by OD<sub>600</sub> measurements), which is close to the theoretical maximum, yet we did not detect any significant alignment of cells based on the circular mean analysis (Figure 6D and Note S5). We therefore conclude that the cell packing is dense but largely disordered, likely similar to a compact disordered spherocylinder packing as was modeled previously (Note S5).

For the material string to rupture under its own weight, all cell-cell contacts in one cross-sectional area need to break apart. Due to the curvature of the cells and the adhesin pair length of \( 2M = 8 \text{ nm} \), only adhesins within a certain contact area can actually contribute to the adhesion between any two cells (Figure 6E and Note S5). We correspondingly define the fraction of the total cell area, \( A_{\text{con}} \) (dimensionless), that is committed to this cell-cell contact and needs to break apart. \( A_{\text{con}} \) depends on how the two cells are placed relative to each other, varying between 0.002 and 0.036 (Figure 6F and Note S5). Furthermore, any cell could have one or multiple such contact areas that need to break apart. This number depends on packing arrangement and, thereby, the total number of nearest neighbors \( N_c \) that any cell has physical contact with, and where many packing arrangements are possible (Figure 6G). Here we consider for simplicity only the ordered packing of a primitive tetragonal lattice, with \( N_c = 6 \), as well as disordered packing, with \( N_c \sim 4 \) (Figure 6G and Note S5). In these cases, only about one such contact area needs to break
Figure 6. Macroscopic material properties can be quantitatively predicted from the molecular and cellular adhesin properties as demonstrated on tensile strength as an example.

(A) Schematic of three snapshots in time of an E. coli cell pellet being extruded from the syringe, where the stress near the syringe increases as the string gets longer.

(B) Picture of the material string breaking.

(C) The rupture length and corresponding tensile strength, $\sigma_{\text{exp}}$, of the material string depending on different adhesion levels (red and green points: different experimental days, whisker plot, min, lower quartile, median, upper quartile, max; circled green dot, outlier).

(D) Confocal image revealing dense packing but no obvious alignment of cells inside the material string (see Note S5).

(E) Schematic of presumed disordered dense packing of the extruded material (top), where due to curvature and size of adhesion molecules two cells can only have a small surface area of connection (red, bottom); this area depends on the relative configuration between cells as shown in (F).

(F) Examples of different pairwise configurations of two attached E. coli cells and the corresponding connection area available for synthetic adhesin binding, $A_{\text{conn}}$ (red area, dimensionless, as fraction of total cell surface area; minimal and maximal possible value on top and bottom, respectively).

(G) Examples of different possible 3D packings of cells inside the material and the corresponding number of contact points, $N_c$: (from top) primitive tetragonal, hexagonal, disordered packing (many other regular packings are possible, with up to $N_c = 12$). Red areas indicate contact points to nearest cells where cell makes physical contact through adhesins.

(H) Excess tensile strength of the material string due to adhesion, $\delta(C)$, at 100 ng/mL: modeling a primitive tetragonal lattice with the long axis of cells arranged perpendicular (blue) and parallel (orange) to the rupture plane, respectively, and modeling disordered packing similar to illustration in (E) (gray) (ordering indicated by h); experimental results from (C) (blue horizontal line). Data points represent mean ± SEM.

Scale bars, 1 cm (B) and 5 μm (D).
per cell pair, and no additional correction factor is required in the following derivation (Figures 6F and 6G; Note S5).

The number of adhesin pairs that need to break per cell is then the product of this area $A_{\text{con}}$ (Figure 6F) and the total number of adhesins per cell $N_{\text{adh}}(C)$ at a given inducer concentration $C$ (Note S5). The homophilic cells encode the Nb and Ag adhesins on a medium- and low-copy-number plasmid with 20–30 and 10–12 copies, respectively. Hence there are about half as many Ag than Nb adhesins, and the Ag adhesins will be the limiting factor for the number of pairwise connections between any two cells that can be formed. We therefore reduce $N_{\text{adh}}(C)$ as determined above for the medium-copy-number plasmid (Figure 2) by a correction factor of $f_p = 0.5$. For $C = 100$ ng/mL aTc induction, this leads to the average number of adhesin pairs connecting any two cells ($f_p \cdot N_{\text{adh}}(C) \cdot A_{\text{con}}$) of $7 \pm 2$ and $130 \pm 30$ for perfect vertical and horizontal packing, respectively (Figure 6G); for the experimentally observed disordered packing, we then estimate an intermediate value of $30 \pm 5$ (Note S5).

The total number of cells involved in the tensile rupture event also depends on the solid-to-volume fraction of cells to the string volume, $f_{sv}$, the cell volume, $V_{\text{cell}}$, and again how cells are ordered, connected to each other, and arranged relative to the rupture plane. Here $h$ is the height of a cell layer (Figure 6H), with $V_{\text{cell}}/h$ being the effective area a cell takes up in the rupture plane, thereby determining how many cells can be packed into this plane, and with $h_{\text{par}} = h$ and $h_{\text{per}} = 2r$ for the highly ordered cases (Figure 6H and Note S5). Ultimately, $N_{\text{cell}}$ is proportional to the cross-sectional area of the material string, $A_s$, and therefore does not explicitly depend on $A_s$ (Note S5). The predicted excess tensile strength of the material due to the synthetic adhesins then is given by

$$\delta_{\text{model}}(C) = f_p \cdot N_{\text{adh}}(C) \cdot A_{\text{con}} \cdot F_b \cdot f_{sv} \cdot h / V_{\text{cell}}.$$  

(Equation 4)

Based on these considerations, we then compared theoretical and experimental excess tensile strength at $C = 100$ ng/mL aTc induction. For the two extreme cell configurations (Figure 6H, orange and dark blue bars), i.e., where cells are perfectly aligned and packed perpendicular or parallel to the rupture plane, respectively, we determined $\delta_{\text{per}} = 0.17 \pm 0.03$ kPa and $\delta_{\text{par}} = 1.70 \pm 0.25$ kPa (errors determined from uncertainties in the determined biophysical parameters). This also implies that such highly ordered material would be highly anisotropic. In a more realistic disordered packing scenario based on our experimental data and analysis (Figure 6D and Note S5), the cell pellet likely consists of some combination of all possible cell configurations (Figure 6F), leading to an intermediate excess tensile strength of $\delta_{\text{avg}} = 0.50 \pm 0.07$ kPa (Figure 6H [gray bar] and Note S5). These quantitative bottom-up predictions potentially have limitations, such as: (1) material thinning before breaking might lead to a smaller cross-section; (2) additional macroscopic material or ordering defects might exist; or (3) adhesin diffusion might lead to more adhesion pairs per contact point. Nevertheless, when comparing with our experimental results, we find agreement approximately within a factor of $\sim 3$ for the disordered model scenario (Figure 6G), being within the bounds of the two highly ordered, anisotropic model extremes (Figure 6H); agreement also holds true for induction level of 300 ng/mL (Note S5).

**DISCUSSION**

In summary, we demonstrated how to measure the key biophysical parameters of synthetic adhesins using a bacterial Nb-Ag toolbox as an example, and how to then quantitatively predict and tune the macroscopic properties of materials...
engineered with these adhesins. All measured parameters are summarized in Table 1 and Figure 1C, and the specific values will vary based on the specific synthetic adhesin pairs, plasmids, and promoters used. Existing measurements or estimates in related systems (Table 1) provide additional confidence in our findings. The molecular bond-breaking force can also be directly related to the molecular $K_D$. Notably, we did not see variations in diffusivity and localization of the adhesins at the cell poles. In this study, we used three different Nb-Ag pairs, i.e., anti-GFP/GFP (Figures 2, 3, and 4), anti-EPEA/EPEA (Figure 5), and anti-p53TA/p53TA (Figure 6), where the latter two correspond to adhesin pairs 2 and 3, respectively, in Glass and Riedel-Kruse. Each of these adhesin pairs have properties that made them particularly suitable for the type of experiments, i.e., fluorescent labeling via GFP, availability of the Ag as easily modifiable peptide, and the absence of cis interactions on the same cell, respectively. We provided sufficient details and controls that should enable others to perform similar characterizations on existing (synthetic) adhesins as well as newly engineered ones.

Our results revealed multiple surprises regarding how cell-cell adhesins can determine and tune the properties of synthetic as well as natural multicellular bacterial systems. We found that only a very small fraction (~1%) and small absolute number of adhesins actually mediate a connection between any two cells, for example, ~1 and ~30 pairs at 30 and 100 ng/mL induction, respectively (Note S4 and Figure 6). Given their specific and rigid cell shape, each cell has adhesin-mediated contacts with only ~4 neighboring cells inside a disordered material, while for highly ordered packing this number can be much higher (Figure 6G and Note S5). Hence, even if initially bound adhesins should become damaged during cell-cell separation, the cellular binding abilities and strengths would stay essentially unaltered over many cellular binding-unbinding cycles, affecting material-level properties such as viscosity.

We also investigated the viability of the adhesin-expressing cells over time under stressful conditions, as well as the stability and capacity to self-regenerate the ELM (Note S6). We previously demonstrated that these synthetic adhesins are compatible with cell growth and division and that they do not affect cell viability. For further validation, we now additionally monitored the stability of the ELM over time in buffer without nutrients and observed that the material consisting of adhesin-expressing cells was significantly improved compared with non-adhesive cells (Note S6). We also determined that the material degradation timescale is within a factor of 3 compared to the degradation of adhesins (Note S6). In addition, we explored the capacity of the material to self-regenerate and determined that even after several days of exposure to stressful environments (e.g., nutrient deprivation or water

<table>
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<tr>
<th>Parameter</th>
<th>Symbol</th>
<th>Unit</th>
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<tr>
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<td>1,200</td>
<td>900</td>
<td>6,000$^a$</td>
<td>Kalisky et al.</td>
</tr>
<tr>
<td>Degradation rate</td>
<td>$b_{adj}$</td>
<td>1/h</td>
<td>0.05</td>
<td>0.01</td>
<td>0.01</td>
<td>0.13</td>
<td>Maurizi</td>
</tr>
<tr>
<td>Dilution rate</td>
<td>$b_{dil}$</td>
<td>1/h</td>
<td>0.82</td>
<td>0.3</td>
<td>0.1</td>
<td>1.8</td>
<td>Alon</td>
</tr>
<tr>
<td>Bond-breaking force</td>
<td>$F_b$</td>
<td>pN</td>
<td>16.2</td>
<td>0.4</td>
<td>1.8</td>
<td>28–45</td>
<td>Klamecka et al.</td>
</tr>
</tbody>
</table>

See also Equations 1, 2, and 3 for definitions: $C_{1/2} = 85.6\pm6.5$ mg/mL, $n = 1.8\pm0.1, N_{adh,max} = 17,400\pm9,600$ molecules/cell, and $a_{max} = 9,900\pm2,100$ molecules/cell/h. Parameters will vary for specific adhesin pairs (especially $F_b$) and genetic inducers, $N_{adh}$ and $a$ are stated for 100 ng/mL aTc.

$^a$Regarding comparison to literature, a different inducer (and, hence, likely effective different induction levels) was used.
evaporation), the cells were still viable and led to culture growth with no significant
difference compared to wild-type cells grown under normal conditions (Note S6).

Many opportunities exist for tunability from the molecular adhesin level to the
macroscopic material level, well beyond tuning via gene induction level, as demon-
strated here (Figure 6C). For example, binding forces could be lowered by changing
the Nb coding sequence responsible for the binding specificity, and the number
of adhesins binding between cell pairs could be increased by using longer linkers be-
tween membrane and adhesin domain (Figure 6E), by changes in the membrane
mobility, or by changing the aspect ratio of cells. Control over the structural order
of cells inside the material should enable significant tuning of material properties
including anisotropy, as is known from classic inorganic materials (Figure 6H).

The capabilities of synthetic adhesins enable many stand-out features of ELMs as
investigated and demonstrated by multiple research groups. For instance, a pro-
grammable living material assembly utilizing bacterial adhesion was demonstrated
in wearable sensors to detect bioelectrical signals, whereby these sensors showed
self-healing within minutes after stretching. Adhesin-expressing cells can also be
integrated in ELMs capable of responding to environmental cues. This property
was explored by several researchers, for example in the development of living bio-
film-based materials for mercury bioremediation or engineered living glue
capable of autonomous self-repair. These applications showcase the potential of
adhesin-expressing cells in designing ELMs, where quantitative understanding
and tuning of the adhesin properties, as demonstrated here, will further support
these developments.

Finally, we note that the analytical and quantitative bottom-up prediction of material
properties as demonstrated here for tensile strength (Figure 6) is still rare and chal-
 lenging for many ELMs and specific properties. Theories that do exist often do
not provide a reasonable agreement giving non-matching assumptions. Our
data and model indicate that the material-level excess tensile strength under the
presented experimental conditions is simply the product of molecular adhesin
bond-breaking force and number of adhesin pairs between neighboring cells, and
that cell dimension, cell geometry, and packing order have a fundamental impact
(Figures 6E–6H). This bottom-up predictive power (Figure 6H) will also enable new
top-down approaches for measuring molecular parameters, for example, deducing
the molecular adhesin binding force from the material’s tensile strength. We expect
that mechanistic bottom-up theories for many complex living materials might be
challenging to develop and test, and instead pragmatic finite element simulations
and machine-learning methods with high predictive power yet fewer mechanistic in-
sights will play a significant role in future materials engineering. Such experi-
mental and theoretical advancements are key for engineered living materials across
different fields, for example, bioprinting of living sensors or therapeutics. Much
of the outstanding work on ELMs has not yet been paired with quantitative bot-
tom-up predictions or rheological and biological characteristics; we hope that our
results will stimulate more combined experimental-theoretical work of diverse natu-
ral and engineered living materials.

EXPERIMENTAL PROCEDURES

Resource availability

Lead contact

Further information and requests for resources should be directed to the lead con-
tact, Ingmar H. Riedel-Kruse (ingmar@arizona.edu).
Materials availability
This study did not generate new unique reagents.

Data and code availability
All analyzed data are available in the manuscript or the supplemental information. Raw data and modeling scripts are available upon reasonable request.

Plasmids and strains
The MG1655 E. coli strain obtained from the E. coli Genetic Stock Center (CGSC 6300) was used for all experiments in this study. Plasmids were transformed into chemically competent cells following standard protocol. Plasmids were sourced from our earlier work and the copy number was indicated by previously published literature: pDSG339 (pSB3K3_TetR_pTet_Neae2v1_antiGFP) (GenBank: MH_492430), pDSG320 (pSB3K3_TetR_pTet_Neae2v1_antiEPEA) (GenBank: MH_492391), and pDSG321 (pSB3K3_TetR_pTet_Neae2v1_antiP53TA) (GenBank: MH_492393) are medium-copy-number plasmids (20–30), whereas pDSG288 (pSB4A3_TetR_pTet_Neae2v1_P53TA) (GenBank: MH_492378) is a low-copy-number plasmid (10–12).

Number of adhesins and spatial surface distribution
Anti-GFP-expressing E. coli cells were aerobically grown overnight at 37°C in Luria-Bertani (LB) medium supplemented with kanamycin. Membrane adhesin expression was induced by adding different concentrations of aTc. Both the induced cells and the wild-type control were harvested by centrifugation at 3,000 rpm for 5 min and resuspended in a PBS solution with 0.5% BSA for surface treatment. GFP was added to the bacterial cells and left to incubate for 30 min, then spun down and washed with 0.5% BSA/0.1 M PBS solution three times. The cells were then added to a solution of GFP of known concentration, \( c = 365 \text{nM} \). Four microliters of cell culture was spotted on a glass slide.

Confocal fluorescence images were recorded with a Zeiss 700LSM laser scanning confocal microscope equipped with a 10-mW laser. The 488-nm laser line was selected for GFP excitation. The sample was visualized with a 63× oil-immersion objective lens (numerical aperture [NA], 1.4), and the pinhole was opened to 1 μm. Images were recorded by scanning the laser over a 16.8 × 16.8 μm field of view. Images were 188 × 188 pixels, with a scan speed of 17.2 μs per pixel and averaged from two successive scans. Laser intensity during image acquisition was maintained at 2% to minimize photobleaching.

Kinetics of adhesin turnover
To determine the adhesin production rate, the anti-GFP-expressing E. coli cells were grown under the same conditions as before except that adhesin expression was not induced. After 24 h, the bacterial cell culture was back diluted 1:1,000 and incubated for another 4 h to reach the exponential growth phase. They were then harvested by centrifugation at 3,000 rpm for 5 min and transferred to LB medium with 0.1 μg/mL aTc. Samples of 100 μL were then collected every 20 min and spun down, followed by the cells being resuspended in a PBS solution to which recombinant GFP was added. The incubation period was 30 min, after which the cells were washed three times with PBS. Four microliters of cell culture was spotted on a glass slide. For the degradation-rate experiment, the membrane adhesin expression was induced by adding 0.1 μg/mL aTc. The overnight culture was back-diluted and incubated and after 4 h, the cells in exponential growth phase were harvested by centrifugation at 3,000 rpm for 5 min, then incubated with recombinant GFP for 30 min, washed three times with a PBS solution at the same centrifugal settings as above, and
transferred to an agarose pad made of PBS and lacking inducer. Two microliters of cells was spotted on the agarose pad.

Confocal images were recorded by scanning the laser over a field of view that was typically 20.3 × 20.3 μm (224 × 224 pixels). Images were averaged from two successive scans. Laser intensity during image acquisition was maintained at 2% and the pinhole to 1 a.u.

**Diffusion coefficient**
Anti-GFP cells were grown as presented above. Membrane adhesin expression was induced by adding aTc. Prior to measurements, the culture was diluted approximately 1:1,000 into the same medium and grown at 37°C under constant shaking for 4 h. Cephalexin was added to 30 g/mL and the cells were grown for a further 45 min. Cells were harvested by centrifugation at 2,000 rpm (to prevent breakage of the elongated cells) for 5 min and resuspended in a PBS solution. Recombinant GFP was then added to the bacterial cells and left to incubate for 30 min, then washed three times with PBS. Four microliters of cell culture was spotted between a glass slide and a coverslip. During the entire experiment, the stage was heated to 25°C using a climate chamber.

The confocal images were acquired using a 63× oil-immersion objective lens (NA, 1.4) under an incubation temperature of 37°C. Images were recorded by scanning the laser over a field of view that was typically 16.9 × 16.9 μm. The resolution of the images was kept low at 32 × 32 pixels, with a scan speed of 16 μs per pixel to minimize photobleaching. Laser intensity during image acquisition was maintained at 2%. A series of ten prebleach images were acquired, after which the laser was pulsed once at a selected region of the cell with 100% intensity to bleach the area but at a lower scan speed to ensure sufficient bleaching. Postbleach images were recorded at 40-ms intervals for a total of 5 s.

**Adhesin bond-breaking force**
Anti-EPEA-expressing E. coli cells were grown overnight, and adhesin expression was induced by adding different concentrations of aTc. The overnight culture was back diluted and induced, and after 4 h, cells were harvested by centrifugation at 3,000 rpm for 5 min. The cells were then resuspended 1:10 in PBS. Biotin-Ahx-EPEA peptide was synthesized by GenScript at >95% purity (Ahx: aminohexanoic acid linker used to create extra space between the biotin and the protein for efficient access of biotin-binding entities). The lyophilized peptides were resuspended in water, and their concentration was quantified on a NanoDrop One using the A205/31 method. The biotinylated protein was then attached to the streptavidin-coated beads, purchased from Spherotech (catalog no. SVP-15-5), according to the protocol presented in the TechNote 101 by Bangs Laboratories. The concentration of protein added to the microspheres was ensured to be in excess such that the binding capacity of the beads reached the maximum. The final solution of protein-coated beads was at a concentration of 0.5 mg/mL.

A flow chamber consisted of a glass slide presenting two holes connected to plastic tubing with the inner diameter of 0.02 inches and a 24 × 50-mm coverslip, attached together by double-sided tape such that the height of the chamber was 0.1 mm. PLL was injected into the chamber through one of the tubes and left to dry overnight. The cell-PBS solution was then introduced into the flow chamber at 0.01 mL/min until it was filled and then left to incubate for 30 min.
Thereafter, the unbound cells were washed using a PBS solution at a flow rate of 0.01 mL/min. Prior to experiments, the bead solution was diluted 1:20 in PBS and injected into the chamber.

The optical trap was custom built on an inverted Nikon Eclipse TE200 microscope with a Nikon 100 x, 1.4 NA oil-immersion objective and a 10 W Nd:YAG 1,064-nm infrared trapping laser. The trapping beam was steered using an IntraAction Acousto-optic Deflector (AOD). The driving voltage frequency was controlled by an IntraAction frequency board in which the frequency was controlled by custom-written LabView software operated from a Windows 10 operating system. Imaging of the bead and bacteria was completed using a C7300 Hamamatsu camera using custom-written LabView software. The protein-coated microsphere was captured using the optical trap with the laser moving back and forth using an AOD and a custom-written LabView code. The laser frequency of oscillation of $n_{\text{laser}} \sim 0.01 - 0.22$ Hz and a spatial amplitude, $A$, of oscillation were tuned to ensure contact between bead and E. coli and subsequent bond breakage ($A$ was typically between 2.2 and 2.3 $\mu$m). For calibration of the trapping laser spring constant, a spherical polystyrene bead was held at the focal point of the trapping laser while calibration was performed. Using custom-written LabView software, the trapping stiffness, $k_{\text{las}}$, was measured by two methods: equipartition theorem and power spectrum analysis. To obtain this information, a low-power green laser aligned with the trapping laser was used in combination with a Thor Labs Photo-Quadrant Diode (PQD). The calibration laser was focused down to the back focal plane of the objective in which the PQD reads a voltage level associated with the displacement of the trapped bead. Because the trapping stiffness scales linearly with laser power level, to determine the trapping stiffness at full power, a lower power (and therefore lower $k_{\text{las}}$) is typically used in conjunction with equipartition and power spectrum analyses. The measured $k_{\text{las}}$ was then scaled linearly along with the laser’s power level to achieve the desired final trapping stiffness. For the spectral density measurements, the displacement of the calibration bead was transformed into a power spectral density in which the cutoff frequency was used to determine optical trap stiffness.

The flow chambers containing anti-EPEA cells and EPEA-coated microspheres were mounted onto the stage of the optical trap. Manually manipulating the stage, a bead was caught at a distance of approximately 4 $\mu$m above the glass coverslip (i.e., above the bottom of the slide). This bead was then positioned near an E. coli cell that had been adhered to the coverslip. The bead was then lowered to the glass coverslip until contact was made, indicated by observing the bead leaving the focal plane. The bead was then raised away from the coverslip by approximately 200 nm. The position of the stage was then held constant, and the trapping beam was moved by varying the AOD input acoustic frequency ($n_{\text{AOD}}$) so that the bead moved laterally back and forth in a triangular wave pattern at a constant frequency ($n_{\text{laser}}$). The position of the stage was adjusted so that the bead made contact with the bacterium at the peak of the triangular wave pattern. As the laser is moved away from the bacterium, if the bead has attached to the bacterium, the bead will start to become horizontally displaced from the focal point of the trapping laser. We conducted 282 individual pull force trials, moving the bead perpendicular to the long axis of the bacterium. Typically, the force-displacement curve of the optical trap is linear; however, in our setup, at values above 400 nm from the trap equilibrium (40 pN with our laser power), the force-displacement relationship becomes non-linear. Therefore, forces above 40 pN (i.e., corresponding to more than two adhesin pairs being ruptured) are not
included in further analysis so as not to skew the final results. Images and AOD input frequency from each trial were analyzed using custom-written MATLAB code in order to correlate the center of the polystyrene bead to the actual location of the trapping laser. Image analysis was performed using custom-written code in MATLAB. The difference in location between the laser focal point and bead, $\Delta s$, was extracted from the data. The rupture-strength force was then computed, $F_{\text{rup}} = k_{\text{las}} \times \Delta s$, where $F_{\text{rup}}$ is the peptide rupture force.

**Tuning ELM properties: Tensile strength**

A preculture (3 mL) of homophilic p53 TA E. coli strain was grown overnight. Next day, the preculture was transferred to 500 mL of LB supplemented with the corresponding antibiotics and after 4 h of incubation, the adhesin expression was induced by adding aTc. Twenty-four hours after induction, the cells were harvested by centrifugation at 4,000 $\times$ $g$ for 10 min. The supernatant was then removed and the cell pellet transferred in a 10-mL syringe. After sealing the syringe tip and fixing the plunger in place, the cell pellet was centrifuged again at 4,000 $\times$ $g$ for 8 min to remove air bubbles. To extrude the material from the syringe, a syringe pump was turned sideways and the flow was set to 0.2 mL/min. A blunt syringe tip with a diameter of 0.8 mm was attached to the syringe tip.

**SUPPLEMENTAL INFORMATION**

Supplemental information can be found online at https://doi.org/10.1016/j.matt.2024.03.019.

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**AUTHOR CONTRIBUTIONS**

S.A.C.: conceptualization, data curation, formal analysis, investigation (all experiments), methodology, validation, visualization, writing – original draft, writing – review & editing. P.M.R.: data curation, formal analysis, investigation (optical trap experiments), methodology, software, writing – original draft, writing – review & editing. H.K.: conceptualization, methodology, writing – review & editing. C.W.W.: methodology, resources, supervision. I.H.R.-K.: conceptualization, formal analysis, funding acquisition, project administration, resources, supervision, writing – original draft, writing – review & editing. All authors have read and approved the manuscript.

**DECLARATION OF INTERESTS**

The authors declare no competing interests.

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